

COMPUTER SIMULATION OF THE ELECTRIC TRANSPORT PROPERTIES OF THE FeSe MONOLAYER

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The paper deals with the model research of electric transport characteristics of stressed and non-stressed FeSe monolayers. Transmission spectra, current-voltage characteristic (CVC) and differential conductivity spectra of two-dimensional FeSe nanostructure have been calculated within the framework of the density functional theory and non-equilibrium Green's functions (DFT + NEGF). It has been shown that the electrophysical properties depend on the geometry of the sample, the substrate, and the lattice constant. On CVC of non-stressed sample in the range from -1.2 V to -1 and from 1.2 V to 1.4 V, a region of negative differential resistance (NDR) has been observed. NDR is at both signs of the applied voltage due to the symmetry of the nanostructure. d^2I/dV^2 is used to determine the nature of the electron-phonon interaction and the features of quasiparticle tunnelling in stressed and non-stressed samples. The results obtained can be useful for calculating new elements of 2D nanoelectronics.

Keywords: *Current-voltage characteristics, FeSe monolayer, transmission spectra, 2D-nanoelectronics.*

1. INTRODUCTION

Currently, reducing the size of electronic components, which consist of the circuits of all modern chips, is becoming increasingly important [1]–[11]. For further miniaturization, it is necessary to develop new nanomaterials with controlled electrophysical properties [12]. With the discovery of graphene and other two-dimensional atomic crystals, the search for new 2D nanomaterials with unusual electrophysical properties began, as well as the development of various electronic devices based on it [13]–[15]. Various technological methods for synthesizing two-dimensional monoatomic crystals based on elements of the fourth group of the periodic table are already known: germanene [16], silicene [17], borophene [18], stanene [19], phosphorene [20], etc., as well as 2D crystals based on carbides, oxides, chlorides, nitrides, transition metal dichalcogenides, etc. [21]. Recently, intensive work has been carried out on the development of nanoelectronic devices based on 2D nanomaterials: field-effect transistors, sensors, power storage devices, solar cells, etc., so a new direction of nanoelectronics has appeared – 2D-nanoelectronics, the main idea of which is to implement elements of electronic technology using 2D-nanomaterials [22]. It opens up the possibility of producing completely new electronic devices with unusual characteristics.

At this moment, the properties of electronic nanomaterials are modelled by using various computer programs. Determining the transmission properties of nanomaterials and the areas of negative differential resistance (NDR) leads to the creation of electronic components with predefined

physical properties.

In this paper, the electric transport characteristics of the FeSe monolayer are modelled and analysed in the framework of the density functional theory in combination with the method of non-equilibrium Green's functions (DFT + NEGF). The first reason of our interest has been to simulate the electric transport properties of the FeSe monolayer, whose tetragonal structure of the crystal is very simple. It is completely characterised by two lattice parameters [23]. Second, when the temperature changes, FeSe undergoes small structural changes. This change between iron atoms causes the material to exhibit directional-dependent behaviour, such as the increased electrical resistance or conductivity, only in the left-to-right or forward-to-backward direction (nematic) [24].

The discovery of superconductivity in 2D-FeSe at a critical temperature $T_c \sim 65$ K was quite surprising, since in bulk 3D-FeSe the T_c value did not exceed 10 K. A possible reason for this record-breaking increase in the critical temperature of the FeSe monolayer is a sharp increase in the electron concentration due to the presence of oxygen vacancies on the surface of the SrTiO₃ substrate [25]. Note that paper [26] proposes an alternative approach to the deposition of potassium atoms on the surface of the FeSe monolayer to change the electron concentration. These features of the FeSe superconducting monolayer are insufficiently explained as part of the traditional phonon and spin-fluctuation mechanism of electron pairing, which requires the study of its electronic properties.

2. GEOMETRY

The geometry of the FeSe model monolayer is shown in Fig. 1 (a–b). Exploration of electric transport characteristics was carried out with samples that were not-stressed (Fig. 1a), stressed with a substrate (Fig. 1b) and without it. When a monolayer is placed on a substrate, the structure is stressed and the linear dimensions increase.

The lattice constant in the layer of the non-stressed sample is 3.765 \AA , and in

the layer of a stressed sample it is significantly larger and is 3.905 \AA . The length of the electrodes along the C axis of the non-stressed sample is $\sim 7.53 \text{ \AA}$, and the stressed sample is $\sim 7.81 \text{ \AA}$. The length of the monolayer under examination reaches $\sim 50.86 \text{ \AA}$. The distance between the electrodes for a non-stressed and stressed structure is 35.8 \AA and 37.14 \AA , respectively.

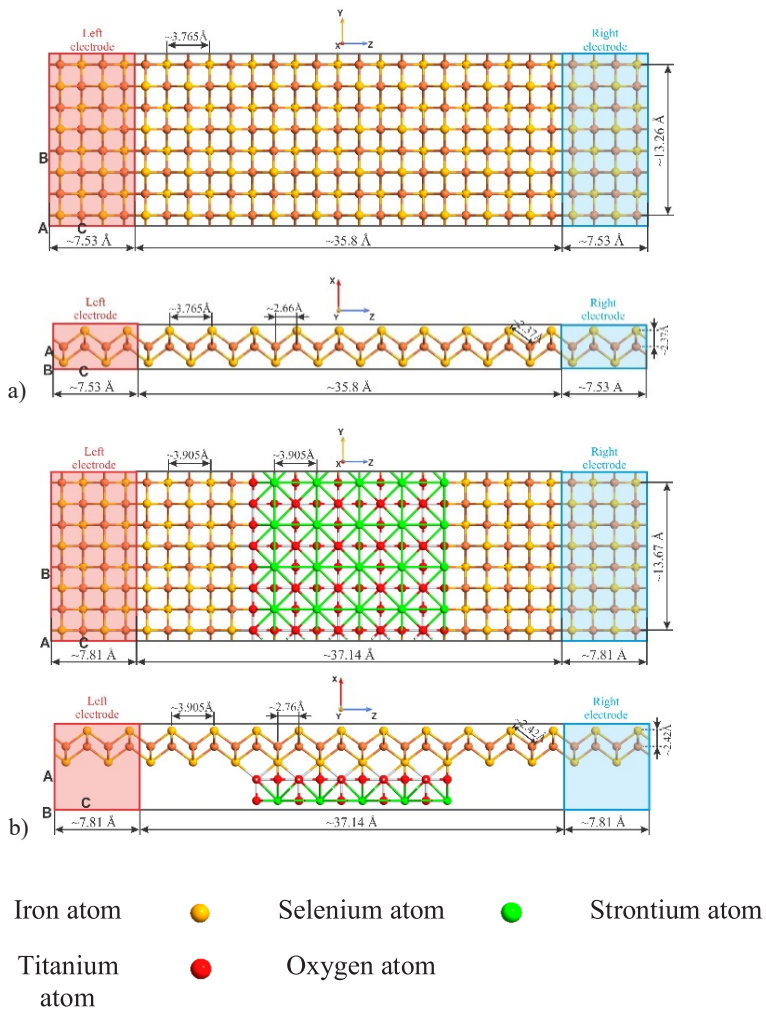


Fig. 1. The geometry of the sample.

For optimization, the 2D-FeSe geometry and description of the interaction between its atoms DFT were used with the

exchange-correlation functional GGA-PBE [27], [28].

3. SIMULATION MODEL AND METHODS

Calculations of electric transport characteristics of 2D-FeSe were performed in the framework of DFT using the method of NEGF in the local density approximation (LDA) [29]. Modelling of quantum transport characteristics of a nanodevice is implemented in the Atomistix ToolKit with

Virtual NanoLab [30]. The main equations of this method are described in detail in our works [31], [32]. The 2D-FeSe CVC is calculated on the basis of the well-known Landauer equation, which indicates the fundamental relationship of the electric current with the transmission spectrum [33]:

$$I(V_L, V_R, T_L, T_R) = \frac{2e}{h} \int_{-\infty}^{+\infty} T(\varepsilon) \left[f\left(\frac{\varepsilon - \mu_R}{k_B T_R}\right) - f\left(\frac{\varepsilon - \mu_L}{k_B T_L}\right) \right] d\varepsilon, \quad (1)$$

where e is the charge of the electron, h is Planck's constant, ε is energy, $T(\varepsilon)$ is function (spectrum) transmission, $f(\varepsilon)$ is the Fermi distribution function of quasiparticle energy, k_B is the Boltzmann's constant, T_R , T_L are the current temperatures and μ_R , μ_L are the electrochemical potentials of the right and left electrode.

The differential conductivity of the FeSe monolayer was obtained from the self-consistent current for a number of applied

displacement and performing numerical differentiation.

4. RESULTS AND DISCUSSION

Evolution of the transmission spectrum, which increases with bias voltage of the monolayer from -2 V to 2 V, is shown in Fig. 2. The bias voltage increased in increments of 0.2 V.

The transmission ability of the FeSe nanostructure increases gradually as the bias voltage grows. The structure of the spectrum gets complicated and new ones appear, preserving the old ones (Fig. 1a). Increase in the number of peaks gives information about increasing indication in the transport of quasiparticles through the structure under consideration [34]. At bias

voltages from 0.4 V and higher (Fig. 2 a), peaks are observed at -2.9 eV, -2.4 eV, -2.2 eV, -1.9 eV, -1.5 eV, -1.2 eV, -0.9 eV, 0.9 eV, 1.1 eV, 1.2 eV, 1.4 eV, 1.6 eV, 1.8 eV, 2 eV, 2.3 eV, 2.6 eV, 2.9 eV. The maximum peak occurred at energies of -1.2 eV and 1.2 eV. A dip in the transmission spectrum was observed at about 0.1 eV. It should be noted that for values from -2 V to -1 V, there is no transmission in the sample. From the transmission spectrum, we can conclude that resonant tunnelling is possible, which is explained by the low dimension of the FeSe monolayer.

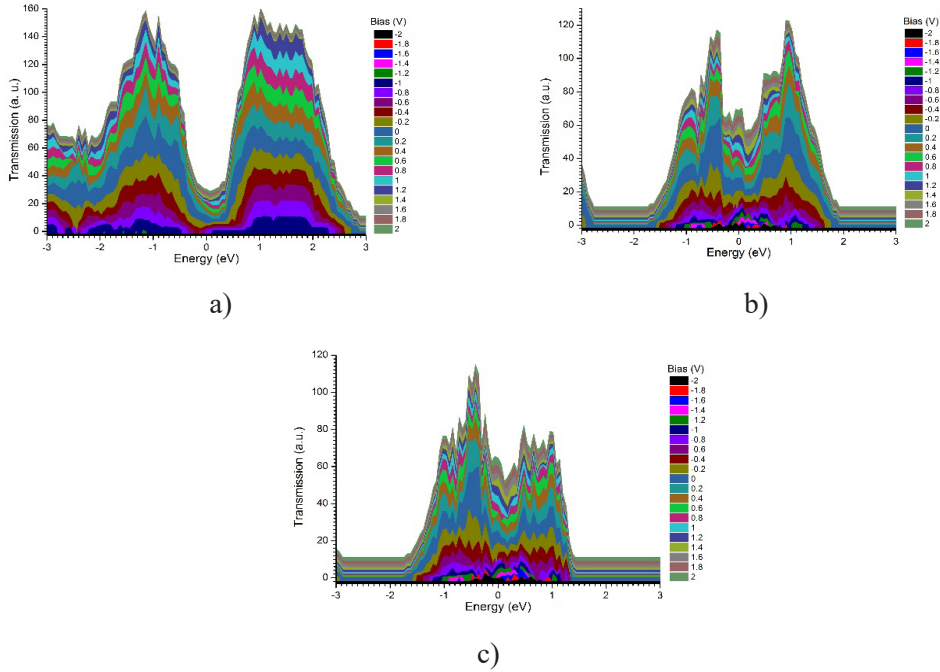


Fig. 2. Evolution of the transmission spectra of a monolayer FeSe:

- a) non-stressed sample;
- b) stressed sample with a substrate;
- c) stressed sample without a substrate.

There are fewer peaks in a stressed sample with a substrate (Fig. 2b), which indicates lower characteristics of the electronic transport of quasiparticles. In this case, the FeSe transmission is recorded from -2 V. The maximum transmission coefficient in this case is less than in a non-stressed sample. The electric transport characteristics are better in a sample with an increased order parameter and without a substrate, judging by the number of peaks, but the interval in which the maxima are observed is more than in a sample with a substrate.

The results of modelling the CVC and differential conductivity are shown in Fig. 3. (The CVC and differential conductivity of the nanostructures under consideration are calculated using Eqs. (1, 2)).

On CVC of the nanostructure, it can be seen that the classical traditional parameter

of order in the voltage ranges from -2 V to -1.2 V, the dependence between current and voltage becomes linear and the current does not flow through the structure. A sharp drop was noted from -1.2 V to -1 V. This is a section of NDR (Fig. 3a (1) and Fig. 3b (1)). The next section of the NDR is marked in the range from 1.2 V to 1.4 V. The minimum was observed at -0.8 V. The maximum value of the non-stressed sample current was $30 \mu\text{A}$. A section of almost linear dependence appeared in the range from 1.6 V, where the current did not flow through the FeSe monolayer. On the curves 2 and 3, negative differential resistance was not observed.

In the samples with an increased order parameter with and without Sr, Ti, O substrate, CVC was almost the same, differing only in the fact that a dip in the nanodevice

with the substrate was at -1.6 eV and CVC became N-shaped. The maximum current of $100 \mu\text{A}$ was also recorded in the sample with the substrate. In both samples, the current increased in the range from -1 V to 1 V from $-100 \mu\text{A}$ to $80 \mu\text{A}$.

In the differential conductivity spectrum in the range from -2 V to 2 V, we observed five peak structures of differential conductivity. Moreover, the intervals of NDR are characteristic only of a non-stressed film. In sample 1 (Fig. 3b) peaks were observed at -1.3 V, -0.7 V, 0 V, 1.1 V, 1.8 V. In the sample, peak structures of $10 \mu\text{S}$, $70 \mu\text{S}$, $90 \mu\text{S}$, $290 \mu\text{S}$ were recorded in the region of bias voltages of -1.5 V, -0.8 V, -0.6 V, 0 V,

0.5 V, 0.9 V, 1.5 V. There are a few maxima in sample 3, but they have higher differential conductivity. The maximum value of the differential conductivity in the negative side was $-230 \mu\text{S}$, in the positive $290 \mu\text{S}$.

Minima were observed around -1.1 V, -1.7 V, 0.9 V, 1.3 V, 1.9 V. The registration of negative differential resistance at both signs of the applied voltage indicates a symmetrical structure of the FeSe molecule [34]. The value of ΔV varied from 0.2 V to 0.7 V. The nature of the resulting maxima on the differential conductivity spectrum is related to the Coulomb interaction of particles.

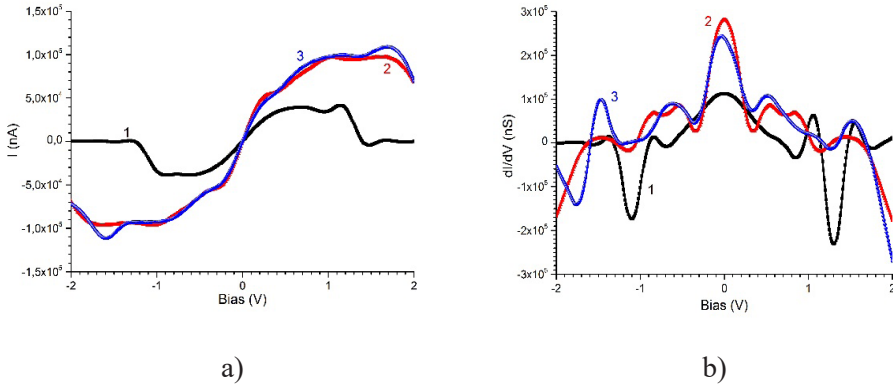


Fig. 3. IV (a) and dI/dV -characteristics (b) of the FeSe monolayer (1 – non-stressed sample, 2 – stressed sample with a substrate, 3 – stressed sample without a substrate).

The graph clearly shows that in a stressed monolayer without a substrate, electronic transport is faster, since the stress period in it is less than in a stressed sample with a substrate.

Figure 4 shows the dependence of the second derivative (d^2I/dV^2) on the bias voltage. It characterises the features of electron-phonon interaction in the FeSe monolayer with different geometric structures, the influence of substrates on the properties of the sample, and the nature of quasiparticle tunnelling.

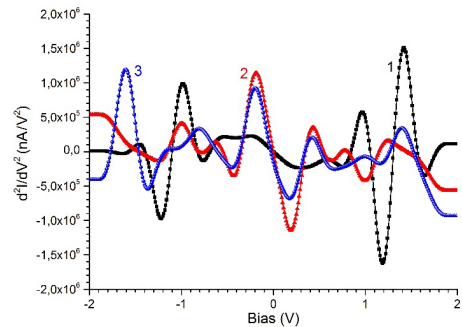


Fig. 4. d^2I/dV^2 - characteristics (b) of the FeSe monolayer (1 – non-stressed sample, 2 – stressed sample with a substrate, 3 – stressed sample without a substrate).

Three distinct peaks were observed at -1 V, 1 V, and 1.5 V in a non-stressed sample (Fig. 4 (1)), relevant to the tunneling conductivity of the FeSe sample. In the background structure of non-stressed sample, the background responsible for the electron-phonon interaction was observed at $-0.8 \text{ V} \div 0.8 \text{ V}$. The dips at -1.2 and 1.2 V (Fig. 4 (1)) are associated with phonon processes and can be compared with the peaks of the fluctuation spectrum of the structure. The most distinct peaks relevant to tunnelling occurred in a stressed sample

without a substrate. The depth of the minimum at 0.2 V in a stressed sample with a substrate was $-12 \mu\text{A}/\text{V}^2$ (Fig. 3(2)) and in a stressed sample without a substrate it was $-0.6 \mu\text{A}/\text{V}^2$ (Fig. 3(3)). Thus, the local minimum was enhanced in the sample when exposed to the substrate in the region of 0.2 V. In stressed samples, the electron-phonon mechanism of tunnelling conductivity prevailed and there were more peaks in the region of -1.8 V , $-1 \div 0.7 \text{ V}$, 0.2 V, 0.5 V, 1.2 V \div 1.4 V.

5. CONCLUSIONS

In this paper, the electric transport properties of the FeSe monolayer have been modelled in accordance with DFT + NEGF. Transmission spectra CVC, dI/dV , d^2I/dV^2 have been determined. It has been shown that the transmission of the nanostructure depends on the geometry of the sample and that the current flows well through the stressed film in contrast to the non-stressed one. The nature of sharp peaks in the transmission spectrum is related to resonant tunnelling of quasiparticles. On CVC of non-stressed sample, a section of NDR is observed in the range from -1.2 V to -1 and from 1.2 V to 1.4 V. NDR is shown

at both signs of the applied voltage, which allows judging about the symmetry of the structure. Based on the second derivative of CVC, we can conclude that the phonon conduction mechanism is manifested in the non-stressed sample. There are three distinct maxima and two minima. It has been shown that the electron-phonon interaction is stronger in a stressed FeSe film than in the non-stressed, and that the frequency of local fluctuation increases when exposed to the SrTiO₃ substrate. The results obtained can be useful for the calculation of new promising electronic devices in 2D nano-electronics.

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